



# High yield of H<sub>2</sub>O<sub>2</sub> and efficient S recovery from toxic H<sub>2</sub>S splitting through a self-driven photoelectrocatalytic system with a microporous GDE cathode

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## ABSTRACT

In this paper, the toxic H<sub>2</sub>S was transformed efficiently into valuable S and H<sub>2</sub>O<sub>2</sub> and its chemical reaction energy was converted into electric energy through a self-driven photoelectrocatalytic (PEC) system, which consisted of a microporous gas diffusion electrode (GDE) and a self-bias photoanode. In the cathode region, high yield of H<sub>2</sub>O<sub>2</sub> up to 0.8 mmol/L/h, was achieved because of the fast gas diffusion and direct two-electron reduction of molecular oxygen on the microporous GDE cathode. In the anodic region, H<sub>2</sub>S was completely converted into S without persulfide based on a I<sup>−</sup>/I<sub>3</sub><sup>−</sup> redox system and the production rate of S recovery was about 0.60 mmol/h, 42 times higher than before. The remarkable catalytic efficiency also benefited from the self-bias design of photoanode, which included a front photoanode of WO<sub>3</sub> and a rear Si photovoltaic cell (Si PVC). The constituted self-driven PEC system could accelerate the charges separation and increase the electron transfer rate of PEC cell. In addition, electricity was generated simultaneously, with a maximum power density of 0.19 mW/cm<sup>2</sup>. The proposed self-driven PEC cell system offers an efficient, complete and sustainable way for H<sub>2</sub>S resourcization.

## 1. Introduction

The industrialization of human society consumes large amount of limited fossil fuels and caused serious environmental problems, threatening the subsistence of all species living on the earth. Seeking efficient approaches to eliminating the pollutants has become a key issue of global concern and many efforts have been made to decompose the pollutants and clean the environment. However, many of pollutants are potentially valuable chemicals rich with energy. For instance, Hydrogen sulfide (H<sub>2</sub>S) is a harmful and undesirable chemical [1–3], while H<sub>2</sub>S is much favorable for dispose to recover elemental sulfur (S) as valuable resources [4]. However, the current processes such as Claus process, consume great amount of energy and will generate other contaminants (SO<sub>x</sub>).

In fact, photocatalytic [5–7] and photoelectrocatalytic (PEC) [8–13] techniques could be a sustainable and cost-effective technology to treat H<sub>2</sub>S [14].



$$\Delta G^\circ = 33.0 \text{ kJ} \cdot \text{mol}^{-1}, \Delta H^\circ = 11.0 \text{ kJ} \cdot \text{mol}^{-1}$$

By introducing the cyclic redox I<sup>−</sup>/I<sub>3</sub><sup>−</sup> system [15,16], the H<sub>2</sub>S can be conveniently and efficiently converted to S and H<sub>2</sub> [17,18]. However, this process was an endothermic reaction and need extra energy [Eq. (1)]. Besides, H<sub>2</sub> requires strict conditions for storage and usage. Comparably, direct conversion of H<sub>2</sub>S into S and H<sub>2</sub>O<sub>2</sub> is more favorable and therefore more easily implemented because this is a spontaneous process on thermodynamics [Eq. (2)]. Moreover, the product H<sub>2</sub>O<sub>2</sub> is an important commodity oxidant [19,20]. How to realize the efficient productin of H<sub>2</sub>O<sub>2</sub> became the key to complete the process.



$$\Delta G^\circ = -106.2 \text{ kJ} \cdot \text{mol}^{-1}, \Delta H^\circ = -151.2 \text{ kJ} \cdot \text{mol}^{-1}$$

To yield H<sub>2</sub>O<sub>2</sub>, an anthra-quinone (AQ) process has been reported [18], in which hydrogenation of AQ with H produces anthrahydroquinone

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(H<sub>2</sub>AQ) and subsequent oxidation of H<sub>2</sub>AQ by O<sub>2</sub> produces H<sub>2</sub>O<sub>2</sub> [21]. However, the production rate of H<sub>2</sub>O<sub>2</sub> was relatively low and the process needed extra addition of chemicals. Actually, the production of H<sub>2</sub>O<sub>2</sub> depends on the reduction of molecular O<sub>2</sub> in the cathodic compartment. Pt, carbon felt and nickel foam are commonly used as cathodes to yield H<sub>2</sub>O<sub>2</sub>, but the output of H<sub>2</sub>O<sub>2</sub> is far from enough. According to previous research, increasing the transfer rate of molecular O<sub>2</sub> and improving the direct two-electron reduction will help to yield H<sub>2</sub>O<sub>2</sub> [22–24]. Based on this consideration, a specially designed gas diffusion electrode (GDE) was chosen as the cathode due to its microporous structure and high specific surface area which could help the diffusion of oxygen molecules to the surface of the cathode and the direct reduction of two-electron process.

On the other hand, how to accelerate the separation of photo-generated charges and increase the electron transfer rate in PEC process is still a key issue. In this study, we also constructed a self-driven PEC cell which consisted of the microporous GDE cathode and a self-bias photoanode to achieve the above objective. Under illumination, the self-driven photoanode could form a self-bias function by its front WO<sub>3</sub> and rear Si photovoltaic cell (Si PVC) [25], in which WO<sub>3</sub> mainly absorbs short-wavelength part of sunlight and generates electron/hole pairs [26–28]. Meanwhile, the filtered long-wavelength light can be captured by rear Si PVC to generate photovoltage [26], which will highly promote the separation of photogenerated charges and electron transfer rate of PEC cell.

In a word, we proposed a self-driven PEC system to decompose toxic H<sub>2</sub>S, recover valuable S and H<sub>2</sub>O<sub>2</sub> and generate electricity simultaneously. This newly designed system achieved remarkable catalytic efficiency, in which the yield of H<sub>2</sub>O<sub>2</sub>, up to 0.8 mmol/L/h, was nearly dozen times as previous report [18], and the production rate of S was remarkably enhanced as well.

## 2. Experimental

Unless otherwise indicated, all the chemicals used in this work were purchased from Sinopharm Group CO. LTD and were of analytical reagent. All solutions were prepared to use high-purity DI water.

### 2.1. Preparation of self-bias photoanode

The self-bias photoanode consisted of a front photoanode of WO<sub>3</sub> and a rear photocathode of Si photovoltaic cell (Si PVC). The effective area of the photoanode was 6 cm<sup>2</sup>.

The WO<sub>3</sub> nano-flake arrays (NFA) was prepared through hydrothermal method [29]. In a typical process, 1 g of ammonium metatungstate((NH<sub>4</sub>)<sub>10</sub>(H<sub>2</sub>W<sub>12</sub>O<sub>42</sub>) · 4H<sub>2</sub>O) was dissolved in 95 mL deionized water and stirred for 1 h. 2 mL of concentrated HCl (37%) was added and stirred for 1 h. Finally, 4 mL of H<sub>2</sub>O<sub>2</sub>(30%) was added and stirred for another 1 h to dissolve the tungstic acid. The WO<sub>3</sub> precursor solution was transferred to a Teflon reactor and the FTO substrate (Sigma-Aldrich, surface resistivity 7.3 Ω cm<sup>-1</sup>, 2.5 × 4.5 cm<sup>2</sup>) was diagonally placed in the reactor. The reactor was then put into the oven and the hydrothermal synthesis was conducted at 160°C for 4 h. After the reactor cooled to the room temperature, the sample was taken out and flushed slightly with DI water. Finally, the sample was calcined in a muffle furnace at 500°C for 2 h with a heating rate of 1°C/min.

The self-bias photoanode composed of a WO<sub>3</sub> NFA electrode, as the front photoanode and a commercial tetra-cell Si cell, as the rear PVC (Figure S1). The two electrodes were connected in series and were sealed by epoxy resin.

### 2.2. The preparation of GDE cathode

Figure S2 shows the fabrication process of GDE. Typically, 2.5 g prepared graphite powder, 0.25 g Na<sub>2</sub>SO<sub>4</sub>, 0.2 g acetylene black and 12.5 mL absolute ethyl alcohol were mixed and shaken in the ultrasonic

machine for 15 min. Then 0.833 g Polytetrafluoroethylene (PTFE) was added into the mixture at 80°C until a paste was formed.

Due to its microporous structure and high specific surface area helps the rapid diffusion of oxygen molecules to the surface of the cathode and the direct reduction of two-electron process. The paste was then coating on the pretreated nickel foam, which was more stable than stainless steel and had a microporous structure to increase the contact area. Then the fashioned electrode was placed in the mold and pressed by a powder pressing machine at 15 Mpa for 30 min. After molding, the electrode was immersed in water at 60°C for 3 h and in acetone at room temperature for 10 h. Finally, the electrode was calcined in a muffle furnace at 300°C for 30 min with a heating rate of 3°C/min. To obtain GDE of different graphite-PTFE mass ratios, the addition amount of PTFE could be changed to 2.5 g (1:1), 1.25 g (1:2) and 0.625 g (1:4).

### 2.3. Structural characterization

The morphology of the samples was characterized by Field emission scanning electron microscopy (FE-SEM) (FE-SEM; Zeiss, Germany, ULTRA PLUS). The crystalline phases and crystallite size of the samples was characterized with X-ray diffractometry (XRD) (Bruker, Germany, AXS-8 Advance) at a scan rate of 5° min<sup>-1</sup>. The BET surface area and the contact angle of the samples were characterized by Drop Shape Analyzer (KRUSS, Germany, DSA100).

### 2.4. Photoelectrochemical measurements

The photo response of the WO<sub>3</sub> NFA photoanode was carried out using a three-electrode system with the samples as the working electrode, platinum as the counter electrode and Ag/AgCl electrode as the reference electrode. The working electrode potential and current were controlled by an electrochemical workstation (CHI 660c, CH Instruments Inc. USA). A 350 W Xe lamp was used as artificial light source, without further description, all experiments were carried out under visible light irradiation (light intensity, 100 mW cm<sup>-2</sup>). The electrolyte was a 0.1 M Na<sub>2</sub>SO<sub>4</sub> solution. The linear sweep voltammograms (LSV) were carried out under chopped light irradiation. The scan rate for the linear sweep voltammetry was 10 mV s<sup>-1</sup>.

### 2.5. The self-driven PEC system for H<sub>2</sub>S splitting

The self-driven PEC system for H<sub>2</sub>S splitting was based on a custom-built double-chamber reactor with a proton membrane (Bofeilai Technology Co., Ltd.) and a corollary equipment for S collection. The self-bias photoanode and GDE cathode were put in the anodic and cathodic compartment, respectively. The electrolytes in the anodic and cathodic compartments were 100 mL 0.5 M H<sub>2</sub>SO<sub>4</sub> containing 0.2 mol/L of KI and 100 mL 0.5 M H<sub>2</sub>SO<sub>4</sub> respectively. Under illumination, H<sub>2</sub>S was periodically pumped into the anodic compartment following a filtration through a corollary S collection device and the filtered solution was injected back to the anodic compartment. H<sub>2</sub>O<sub>2</sub> was generated in the cathodic cell and the concentration was determined through potassium titanium oxalate spectrophotometry by a UV-VIS spectrophotometer (TU1810, Universal Analysis, Beijing, China) at its maximum absorption wavelength of 410 nm [30].

## 3. Results and discussion

### 3.1. Characterization of the electrodes

The top-view SEM image of the WO<sub>3</sub> NFA was shown in Fig. 1 (a). It can be seen that the WO<sub>3</sub> film had a plate-like morphology and the nanoplates adhered tightly and uniformly on the FTO substrate. The FTO substrate was totally covered by the WO<sub>3</sub> nanoplates without any empty region. The WO<sub>3</sub> nanoplates were directly grown on FTO in one step with no seed layer. This direct growth method could avoid the

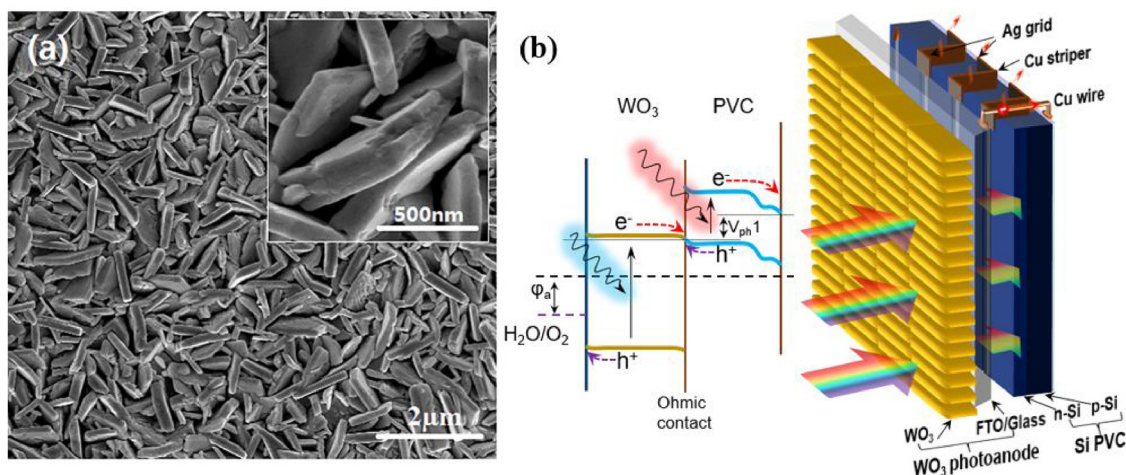


Fig. 1. (a) SEM image of the WO<sub>3</sub> NFA; (b) The schematic diagram of the self-bias photoanode of WO<sub>3</sub> NFA/SiPVC.

grain boundaries of the seed layer, which would reduce the resistance between the photocatalyst and the FTO substrate and thus improve the PEC performance. Previous research showed that the addition of a seed layer or a structure-directing agent resulted in an asymmetrical or discrete distribution of WO<sub>3</sub> NFA on the FTO [31–34]. Such an asymmetrical or discrete distribution of WO<sub>3</sub> NFA wasn't observed in our samples. As previously reported [35], the WO<sub>3</sub> NFA was of orthorhombic phase and the absorption edge lies at 450 nm (Figure S3). The prepared WO<sub>3</sub> NFA has good stability (Figure S4(a)) and photocatalytic capacity under solar light irradiation (Figure S4(b)). It also exhibited incident-photon-to-charge conversion efficiency (IPCE) behavior in the range of 350–470 nm (Figure S4(c)), which is roughly consistent with the light absorption spectrum of the WO<sub>3</sub> NFA. This means that the conversion of the absorbed photons into photocurrent occurred successfully in the photoanodes.

Fig. 1 (b) shows the schematic diagram of the self-bias photoanode with WO<sub>3</sub> NFA as the front photoanode material and a commercial Si cell as the rear photovoltaic cell. The particular structure of Si PVC was shown in Figure S1. The WO<sub>3</sub> and Si PVC are sealed together as a monolithic photoelectrode to simultaneously maximize light utilization and minimize complexity of device fabrication. The Si PVC was adhered to the back of the WO<sub>3</sub> NFA and sealed by silicone rubber. The front WO<sub>3</sub> NFA mainly absorbs short-wavelength part of sunlight and generates electron/hole pairs. The long-wavelength light irradiates through the WO<sub>3</sub> NFA and will be captured by the rear Si PVC to generate additional bias (Figure S5). The photogenerated holes of WO<sub>3</sub> NFA transfer to the electrode/electrolyte interface, meanwhile the electrons get accelerated via the rear Si PVC. Due to the bias by Si PVC, the WO<sub>3</sub> NFA/SiPVC photoanode could be self-sustaining in the PEC H<sub>2</sub>S splitting.

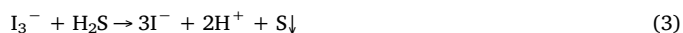
The schematic diagram of the GDE was shown in Fig. 2 (a). Two pieces of graphite and one piece of nickel foam sandwiched between them made up the H<sub>2</sub>O<sub>2</sub>-generating GDE. Fig. 2 (c) shows the SEM image of the nickel foam, which acts as the conducting material of the GDE. It can be observed that the nickel foam has highly-developed hole structure and large specific surface area, which will allow more molecular O<sub>2</sub> to contact the electrode. Besides, nickel was relatively stable. These qualities made nickel foam a better choice than stainless steel, which was used as the conducting material in previous research [36]. The surface SEM of GDE were shown in Fig. 2 (d). It can be observed that the surface of the electrode was loose and porous. With the BET surface area of 8.5766 m<sup>2</sup>/g, the porous structure of GDE could benefit the delivery of O<sub>2</sub> and promote the generation of H<sub>2</sub>O<sub>2</sub>. The PTFE was added during the preparation of GDE to bond the graphite powder. After the electrode took shape, part of the PTFE would be leached by acetone, which could help to shape the porous structure of the

electrode. The surface of the GDE was strongly hydrophobic, with a contact angle of 115.0° (Fig. 2(b)), which could enhance the transfer of O<sub>2</sub> and increase the output of H<sub>2</sub>O<sub>2</sub>. The reaction pathway of oxygen reduction has been proved by LSV tests and Koutechy-Levich (K–L) calculation. As shown in Figure S4(d), the GDE cathode has excellent selectivity of two-electron reduction of molecular oxygen.

### 3.2. Photoelectrocatalytic splitting of H<sub>2</sub>S

The WO<sub>3</sub>/Si PVC photoanode and GDE cathode was applied in the PEC conversion of H<sub>2</sub>S into elemental S and H<sub>2</sub>O<sub>2</sub>. As known, direct oxidation of S<sup>2-</sup> using most semiconductor photoanode commonly leads to the formation of polysulfide (S<sub>x</sub><sup>n-</sup>) by hydroxyl radical (E° = 3.2 V vs. NHE) or photoholes (E° = 2.81 V vs. NHE) because of their high oxidation potential [37], thus a typical yellow clarifying solution was only obtained (Figure S6). The polysulfide can be further confirmed by a qualitative analysis of HCl, in which the milk white turbidity was observed after dilute HCl was added into the yellow clarifying solution. The results indicated that H<sub>2</sub>S could be easily turned into polysulfide (S<sub>x</sub><sup>n-</sup>). In this case, the promoter has to be introduced into the PEC system that makes the S<sup>2-</sup> totally converted into S. I<sup>-</sup> ion is well-known hole trapping agent which can greatly enhance the separation of electron-hole pair and increase the photoconversion efficiency. In the PEC system [37,38], I<sup>-</sup> ions could be first oxidized into I<sub>3</sub><sup>-</sup> ions by WO<sub>3</sub> and thereafter H<sub>2</sub>S could be oxidized into S by I<sub>3</sub><sup>-</sup> ions which was then reduced to I<sup>-</sup> again for the next cycle. Based on the principle, the device configuration and the operation mechanism of the PEC splitting H<sub>2</sub>S system was shown in Fig. 3 (a) in which the WO<sub>3</sub>/SiPVC photoanode and GDE cathode were placed in the anodic compartment and cathodic compartment respectively, separated by a PEM (Proton Exchange Membrane). During operation, H<sub>2</sub>S was injected into the anodic compartment and elemental S was collected through an accessorial device. In the meantime, H<sup>+</sup> was transformed into H<sub>2</sub>O<sub>2</sub> in the cathode region. The reactions (Eqs. (1)–(4)) were shown as follows:

Anodic compartment:



Cathodic compartment:



Fig. 3b shows the open-circuit voltage (V<sub>OC</sub>) of the PEC system under chopped illumination. As can be seen, the V<sub>OC</sub> values were −0.22 V



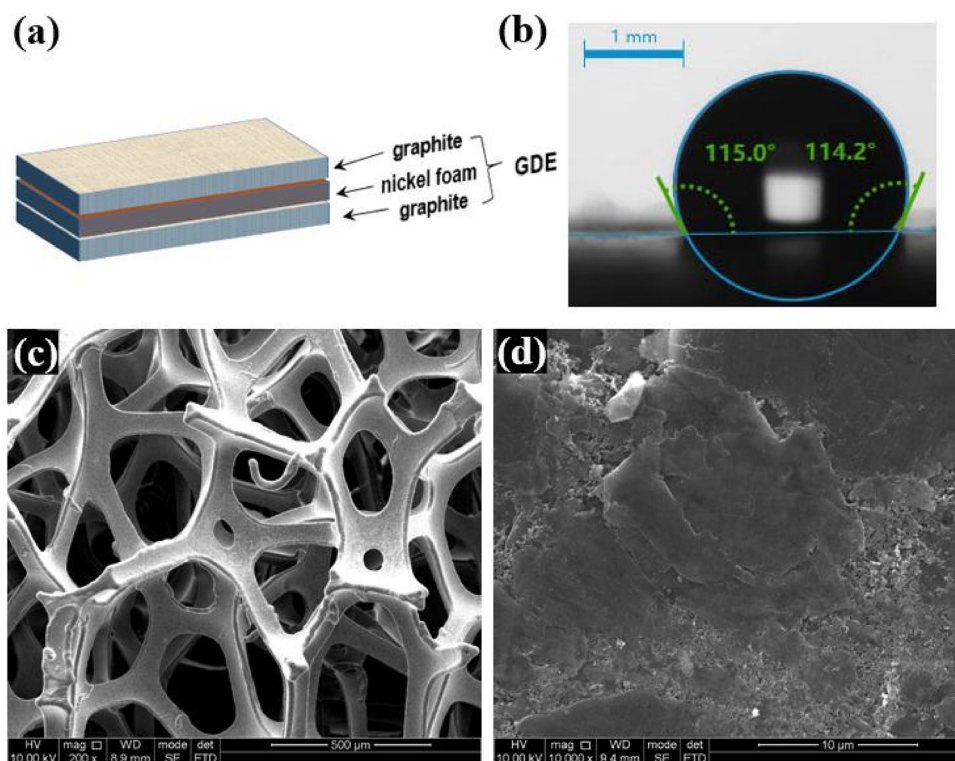


Fig. 2. (a) The schematic diagram of the GDE; (b) The contact angle of the GDE; (c) The SEM image of the nickel foam and (d) the surface of the GDE.

under dark and rose to 2.1 V under illumination. This is according with the photovoltage generated by the rear Si PVC ( $V_{ph} = V_{ph1} + V_{ph2} + V_{ph3} + V_{ph4}$ ) (Figure S7). This result indicates that the PEC system can be driven by domestic bias.

The  $H_2S$  splitting was carried out under visible light illumination when  $H_2S$  was periodically pumped into the anodic compartment. Fig. 4 (a) shows the variation of photocurrent and the relevant color change of the solution in the whole process. In the initial phase, the photocurrent was about 14 mA and the solution color of the anodic compartment gradually changed from clear and colorless to canary yellow and to vivid red, which indicated the generation of  $I_3^-$ . After the injection of gas  $H_2S$ , the photocurrent suddenly fell to about 2 mA while the vivid red solution changed into a faint yellow turbidity simultaneously. This illustrated the formation of elemental S which was nearly insoluble in

water. A clear and colorless solution was achieved again after the turbidity was filtered out. Simultaneously, the photocurrent returned to 14 mA. This process could be constantly repeated in a long term and abundant elemental S would be collected. S production won't influence the performance of  $WO_3$  and it can be reused (Figure S8). In the whole process, the photocurrent was kept at about 14 mA except when gas  $H_2S$  was injected into the anodic compartment. This process can be confirmed by investigating the ion ratio of  $I^-/I_3^-$  during the operation. It was found from Figure S9 that  $I^-$  ions gradually converted to  $I_3^-$  ions in the initial operation of 1000s. When  $H_2S$  was injected in the anodic compartment,  $I_3^-$  ions react with  $H_2S$  and was converted into  $I^-$  ions again and S was generated in the meantime. After that, the system entered a second cycle (Figure S9) and the concentrations of  $I^-$  and  $I_3^-$  exhibited a dynamic equilibrium. The production rate of S through the

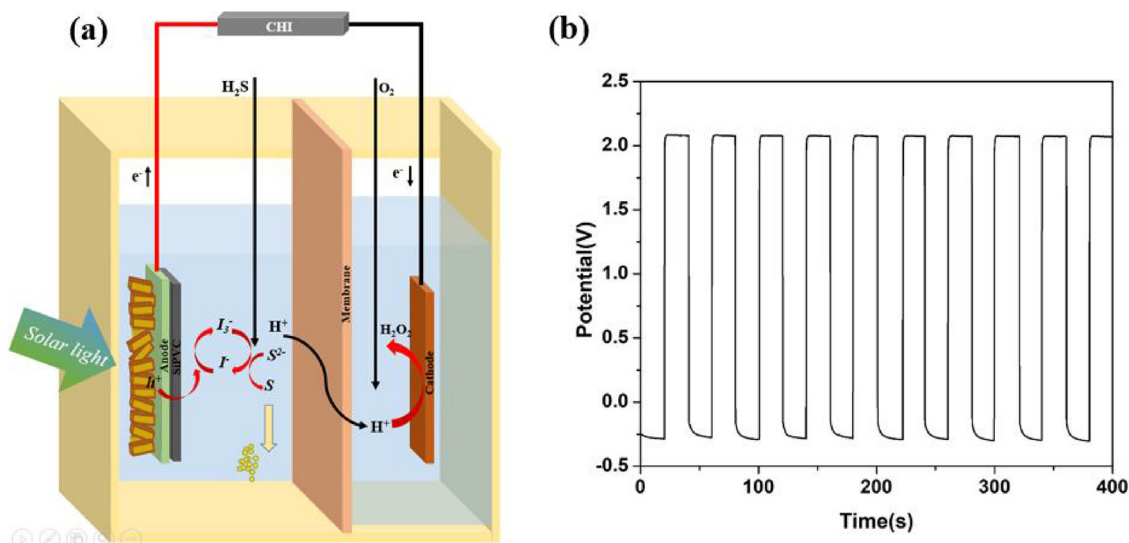


Fig. 3. (a) The schematic diagram of the PEC system; (b)  $V_{oc}$  of the PEC- $H_2S$  system.

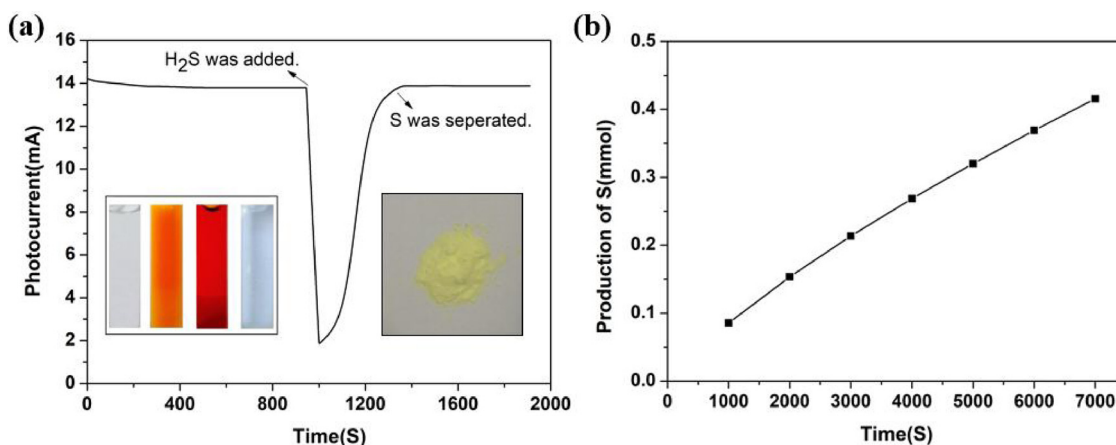


Fig. 4. (a) The current and solution color changes before and after the injection of H<sub>2</sub>S; (b) The S recovery during a long-term operation.

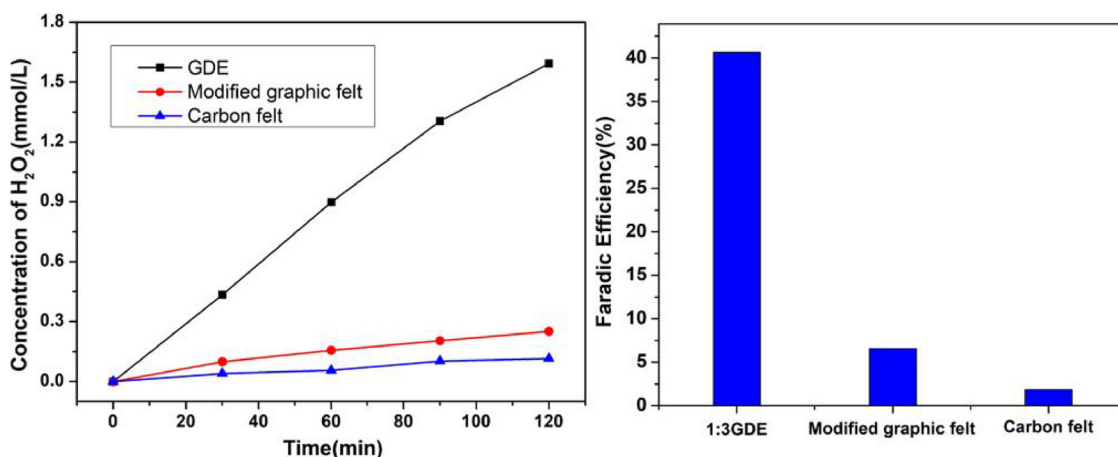


Fig. 5. (a) Production of H<sub>2</sub>O<sub>2</sub> and (b) Faradic efficiency by different cathodes in the PEC system.

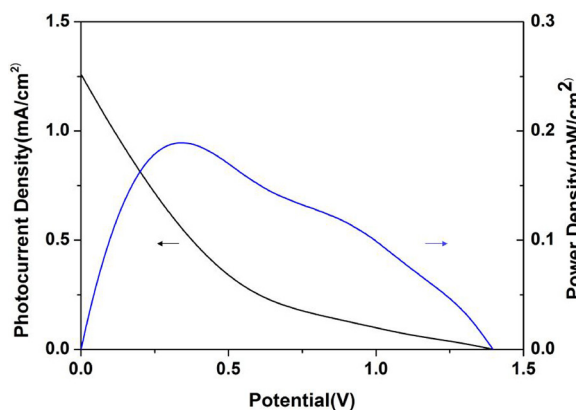


Fig. 6. I–V characteristic curve and power density curve of the PEC-H<sub>2</sub>S system.

PEC system was shown in Fig. 4(b). It can be seen that the production rate of S was maintained at about 6.816 mg h<sup>−1</sup>, respectively, and no polysulfide appeared, suggesting that the S<sup>2−</sup> was totally converted into S.

### 3.3. H<sub>2</sub>O<sub>2</sub> production

Fig. 5 (a) shows the concentrations of H<sub>2</sub>O<sub>2</sub> generated by different photocathodes. The highest rate of 0.80 mmol/L/h was obtained by GDE cathode, which is 6 and 13 times larger than those of modified graphic felt and normal carbon felt cathode respectively. This contributes to the distinctive structure of the GDE as previously mentioned. In previous research, H<sub>2</sub>O<sub>2</sub> was generated through an anthraquinone (AQ) process. The hydrogenation of AQ with H<sub>2</sub> produces anthrahydroquinone (H<sub>2</sub>AQ) and subsequent oxidation of H<sub>2</sub>AQ by O<sub>2</sub> produces H<sub>2</sub>O<sub>2</sub> [21]. AQ process was far more complicated and energy-consuming. On contrast, our process just needed aeration and no medicine was added [18]. The Faradic efficiency was also calculated according to the following formula:

Table 1

The production of S and H<sub>2</sub>O<sub>2</sub> and electricity generation during the first three cycles.

Cycle	1		2		3	
Time(S)	0–1000	1000–2000	2000–3000	3000–4000	4000–5000	5000–6000
S(mmol)	0.086	0.154	0.214	0.269	0.320	0.369
H <sub>2</sub> O <sub>2</sub> (mmol)	0.023	0.046	0.070	0.093	0.116	0.139
Electricity generation(mWh)	0.317	0.633	0.950	1.27	1.58	1.90

$$FE = \frac{nFc_{H_2O_2}V}{\int_0^t Idt} \times 100\%$$

The  $n$  represents the electron transfer number.  $F$  represents Faraday constant.  $c_{H_2O_2}$  represents the concentration of  $H_2O_2$ .  $V$  represents the volume of the solution. As shown in Fig. 5 (b), the Faradic efficiency of GDE was 40.7%, which was 21 and 67 times higher than those of carbon felt and Pt cathode respectively.

The PTFE: graphite mass ratio PTFE is crucial to the performance of the GDE cathode. PTFE serves two purposes: One is bonding the graphite and other materials to form a tough electrode. The other is strengthening the hydrophobicity property of the electrode. However, overmuch PTFE will reduce the porosity of the GDE and influence the  $O_2$  transfer. Besides, it is an insulating material so excess PTFE will increase the electrode's resistance and degrade its electrochemical performance. Therefore, an optimal addition amount of PTFE should be determined. In this study, the best GDE was made by the PTFE: graphite mass ratio of 1:3 (Figure S10).

### 3.4. Electricity generation

As shown in Fig. 6, the electricity generation during the process was studied by assessing the output power density. The decomposition of  $H_2S$  into  $S$  and  $H_2O_2$  was an exothermic process, in which the reactions are beneficial to the electricity generation by the photoanode. First, photogenerated electrons and holes were generated under illumination in the anodic compartment.  $I^-$  could rapidly trap the photogenerated holes and the photogenerated electrons were delivered to the cathodic compartment. This process highly promoted the charge separation efficiency. On the other hand, the rear PVC could capture the filtered long-wavelength light which improved the light utilization and mainly contributed the electricity generation. As shown in Fig. 6, the system's maximum power density was about  $0.19 \text{ mW/cm}^2$ .

To assess the performance of the PEC system, the yields of  $S$  and  $H_2O_2$  were continuously recorded within a long-term operation. Table 1 shows the yields of  $S$  and  $H_2O_2$  and electricity generation within three cycles. The production rates of  $S$  and  $H_2O_2$  were  $0.60 \text{ mmol/h}$  and  $0.08 \text{ mmol/h}$ , respectively, and electricity generation rate was about  $0.19 \text{ mW/cm}^2$ . The deficiency of  $H_2O_2$  may result from the reduction of  $O_2$  into  $H_2O$ . The results also suggested that  $H_2S$  was entirely converted into  $S$  and no polysulfide was generated.

## 4. Conclusions

In summary, we proposed a self-driven PEC system, with  $WO_3/SiPVC$  as photoanode and GDE as cathode, to convert  $H_2S$  into  $S$  and  $H_2O_2$ , and electricity was generated simultaneously. The performance of the self-driven PEC system was studied by measuring the photoelectric effect and the production of  $S$  and  $H_2O_2$ . The results demonstrated that the production rate of  $S$  was maintained at about  $6.816 \text{ mg h}^{-1}$  and no polysulfide appeared. The concentration of  $H_2O_2$  was much higher than previous research's, with an output of  $0.8 \text{ mmol/L/h}$ . The system's maximum power density was about  $0.19 \text{ mW/cm}^2$ . We hope that the novel PEC system gives its promising in efficient and sustainable way of  $H_2S$  waste recycling and energy recovery.

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## Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.apcatb.2018.07.051>.

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